

# How cations determine the interfacial potential profile: relevance for the CO<sub>2</sub> reduction reaction

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## ABSTRACT

The strong effect of the electrolyte cation on the activity and selectivity of the CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) can only be understood and controlled if the cation's effect on the interfacial potential distribution is known. Using CO (the key intermediate in the CO<sub>2</sub>RR) adsorbed on Pt as a probe molecule, and combining IR spectroscopy, capacitance measurements and ab initio molecular dynamics, we show that the cation size determines the location of the outer Helmholtz plane, whereby smaller cations increase not just the polarisation but, most importantly, the polarizability of adsorbed CO (CO<sub>ad</sub>) and the accumulation of electronic density on the oxygen atom of CO<sub>ad</sub>. This strongly affects its adsorption energy, the degree of hydrogen bonding of interfacial water to CO<sub>ad</sub> and the degree of polarisation of water molecules in the cation's solvation shell, all of which can deeply affect the subsequent steps of the CO<sub>2</sub>RR.

**Keywords:** double-layer structure; cation effects; CO<sub>2</sub> reduction reaction; adsorbed CO; vibrational Stark effect

## 1. Introduction

Electrochemistry is central to a multitude of technologically relevant processes. The essential event in any electrochemical process is the electron transfer across the interface between two heterogeneous media. The microscopic structure of the electric double layer (EDL) plays, therefore, a critical role in determining the rate and selectivity of electrochemical reactions. Particularly, the potential profile across the interface is of the utmost importance, because it can affect the stability of adsorbed intermediates[1–3] or polarise molecules within the EDL [4,5].

The effect of the electrolyte cation on the efficiency and selectivity of the CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR) was discovered by Hori and co-workers [6], although the effect of cations on the kinetics of electrochemical reduction reactions had been known since Frumkin's

pioneering work in the 1930's [7]. The dependence of the location of the outer Helmholtz plane (OHP) on the size of the hydrated cation has been identified as one of its main causes [8–10].

Singh et al. [4] have recently suggested that the  $pK_a$  of cation hydrolysis at the EDL differs from that in the bulk, due to the polarisation of water molecules in the cation's hydration shell, which results in cation-dependent interfacial buffering capacities. As the activity and selectivity of the  $CO_2RR$  are pH dependent [11–13], this could explain the cation effect on this reaction. Our group [5] later provided spectroscopic evidence that the pH increase at the interface during the  $CO_2RR$  in  $CO_2$ -saturated bicarbonate buffers is indeed cation dependent. Pérez-Gallent et al. [14] have also recently shown that adsorbed  $COCOH$  (the likely intermediate in the formation of  $C_2$  products) is better stabilised in the presence of  $Li^+$ ,  $Na^+$  or  $K^+$  than with  $Rb^+$  or  $Cs^+$ . Although differing in the details, the interpretations of both Singh et al. [4] and Pérez-Gallent et al. [14] rest both on cation-induced differences in the electric field at the EDL. An in principle alternative explanation was proposed very recently by Li et al. [15], who suggested that the degree of hydrogen bonding of interfacial water to adsorbed  $CO$  ( $CO_{ad}$ ) is cation dependent, but did not discuss how and why different cations lead to different degrees of hydrogen bonding of interfacial water to surface species. As we will show in this contribution, this is also a consequence of cation-induced differences in the electric field at the EDL.

$CO$ -terminated Pt electrodes are an excellent platform to investigate double layer effects. They have been very well characterised spectroscopically and electrochemically [16–24].  $CO$  adsorbs strongly on Pt, remains in the same structure over a wide potential range, and blocks the specific adsorption of ions on the electrode surface. These characteristics also make it most suitable for first-principles calculations. Furthermore,  $CO_{ad}$  is an intermediate in the reduction of  $CO_2$  to hydrocarbons [25–27], and evidence of a cation effect on its stability or

on its interaction with interfacial water has therefore direct relevance to understanding the CO<sub>2</sub>RR. Recently, Koper and co-workers [23,24] have shown that both the interfacial capacitance and the dependence of the C-O stretching frequency of a CO-terminated Pt electrode on the potential ( $\frac{\partial \bar{\nu}_{\text{CO}_L}}{\partial \Delta\phi}$ , often called the Stark tuning rate, STR) are independent of the concentration of the supporting electrolyte for concentrations above 10<sup>-3</sup> mol L<sup>-1</sup>. This implies that the interfacial capacitance is dominated by that of the Helmholtz layer (so-called Helmholtz capacitance, C<sub>H</sub>), which must be determined by the location of the OHP and, therefore, CO-terminated Pt electrodes offer an ideal platform to explore the consequences of a cation-dependent location of the OHP.

Here, we combine surface-enhanced infrared absorption spectroscopy in the attenuated total reflection mode (ATR-SEIRAS) and capacitance measurements to show that the size of the hydrated cation determines the interfacial potential profile. We develop a theoretical formulation that explains the experimental observations and is supported by ab-initio molecular dynamics (AIMD) simulations, and identify the key effects, all of them due to cation-induced differences in the electric field at the EDL, that combine and reinforce each other to yield the strong cation sensitivity of the CO<sub>2</sub>RR. An adequate control of the delicate balance between these three effects should lead to a better control of the activity and selectivity of this extremely important reaction.

## **2. Experimental**

**2.1. Materials.** The working electrode was a Pt film deposited on the totally reflecting plane of a Si prism bevelled at 60°, following a procedure reported elsewhere [28]. The Pt-covered Si prism was attached to the spectroelectrochemical cell using an O-ring seal. Electrical contact to the film was made by pressing onto it a circular platinum wire. Prior to any IR measurements, the electrode was cycled repetitively in 0.1 M HClO<sub>4</sub>. The cell was then rinsed thoroughly with Ultrapure water (Milli-Q) and fill with the desired electrolyte. A

Ag/AgCl (KCl<sub>sat</sub>) electrode was used as reference, but all the potentials in the text are referred to the SHE. The counter electrode was a flame-annealed Pt wire.

Electrolytes were prepared dissolving HClO<sub>4</sub> (70%, Merck p.a.), H<sub>2</sub>SO<sub>4</sub> (96%, Merck Suprapur), LiClO<sub>4</sub> (Aldrich, 99.99% trace metals basis), NaClO<sub>4</sub> (Fluka for HPLC, ≥ 99.0%), K<sub>2</sub>SO<sub>4</sub> (Aldrich, 99.99% trace metals basis), Rb<sub>2</sub>SO<sub>4</sub> (Aldrich 99.8% trace metals basis), CsClO<sub>4</sub> (Aldrich, 99.995% trace metals basis), MgSO<sub>4</sub>·7H<sub>2</sub>O (Sigma BioUltra ≥ 99.5%), Sr(ClO<sub>4</sub>)<sub>2</sub>·3H<sub>2</sub>O (Alfa Aesar 98%), Ca(ClO<sub>4</sub>)<sub>2</sub>·4H<sub>2</sub>O (Aldrich 99%), (NH<sub>4</sub>)<sub>2</sub>SO<sub>4</sub> (Aldrich 9.999% trace metals basis), [N(CH<sub>3</sub>)<sub>4</sub>]ClO<sub>4</sub> (Acros Organics 98%), [N(CH<sub>2</sub>CH<sub>3</sub>)<sub>4</sub>]BF<sub>4</sub> (Sigma-Aldrich for electrochemical analysis, ≥ 99.0%), [N((CH<sub>2</sub>)<sub>2</sub>CH<sub>3</sub>)<sub>4</sub>]ClO<sub>4</sub> (Acros Organics 98%) or [N((CH<sub>2</sub>)<sub>3</sub>CH<sub>3</sub>)<sub>4</sub>]ClO<sub>4</sub> (Sigma-Aldrich for electrochemical analysis, ≥ 99.0%) in ultrapure water (MilliQ). All solutions were prepared to a concentration 0.1 M except in the case of [N(CH<sub>2</sub>CH<sub>3</sub>)<sub>4</sub>]BF<sub>4</sub>, for which a concentration of 0.01 M was used. In order to confirm that the lower concentration does not affect either  $C_T$  or  $\frac{d\bar{v}}{d\Delta\phi}$  experiments were also performed with a 10<sup>-3</sup> M concentration of [N(CH<sub>2</sub>CH<sub>3</sub>)<sub>4</sub>]BF<sub>4</sub>, and the same values were obtained for both magnitudes. As the potential region of stability of CO<sub>ad</sub> is much more negative than the potential of zero charge (pzc) of a CO-covered Pt electrode (1.10 V vs. SHE for Pt(111)[18]), we do not expect the nature of the anion to have any effect on the spectra or, in general, the interfacial structure and properties. For H<sup>+</sup>, both H<sub>2</sub>SO<sub>4</sub> and HClO<sub>4</sub> yielded the same results.

The electrolyte was saturated with research grade carbon monoxide (99.9995 % purity from BOC) for 30 minutes before every experiment, and CO was bubbled through the electrolyte throughout the experiment.

**2.2. Experimental Methods.** ATR-SEIRA spectra were recorded using a Nicolet iS50R FTIR spectrometer equipped with an MCT detector and a home-made ATR accessory, using unpolarized light. Differential spectra are reported in absorbance units, calculated as

$-\log(R_{\text{sample}}/R_{\text{reference}})$ , where  $R_{\text{reference}}$  and  $R_{\text{sample}}$  are the reference and sample spectra, respectively. Positive bands corresponds to species present (or at higher concentration) in the sample spectrum that were absent (or at lower concentration) in the background spectrum, while negative bands correspond to species present (or at higher concentration) in the background spectrum that are absent (or at lower concentration) in the sample spectrum. Potential-step differential spectra were obtained by accumulating 100 interferograms with a resolution of  $4 \text{ cm}^{-1}$ . The electrode potential was controlled with an EmStat3 mini USB potentiostat (PalmSens).

The interfacial capacitance of a CO-covered Pt electrode in the different electrolytes was recorded by cycling the potential of a polycrystalline Pt electrode between 0.1 and 0.4 V at  $0.05 \text{ V s}^{-1}$ . The capacitance was obtained by dividing the current density (which is constant within this potential range) by the scan rate.

**2.3. Computational methods.** The CO-covered Pt(111) surface was modelled using a  $p(4 \times 4)$  periodic slab of 4 layers of Pt atoms. Based on the experimental results, the total surface CO coverage is set to be  $3/4 \text{ ML}$  ( $2 \times 2$  arrangement), with  $1/4 \text{ ML}$  CO on top sites and  $1/2 \text{ ML}$  on three-fold hollow sites. The box size of the simulation model is  $11.246 \times 11.246 \times 40 \text{ \AA}^3$ , and three dimensional periodic boundary condition is applied. Then, the vacuum space between the geometrically-optimized Pt(111)-CO surface and its images is fully filled with water molecules. This model replicates the PZC condition of the Pt(111)-CO/water interface.

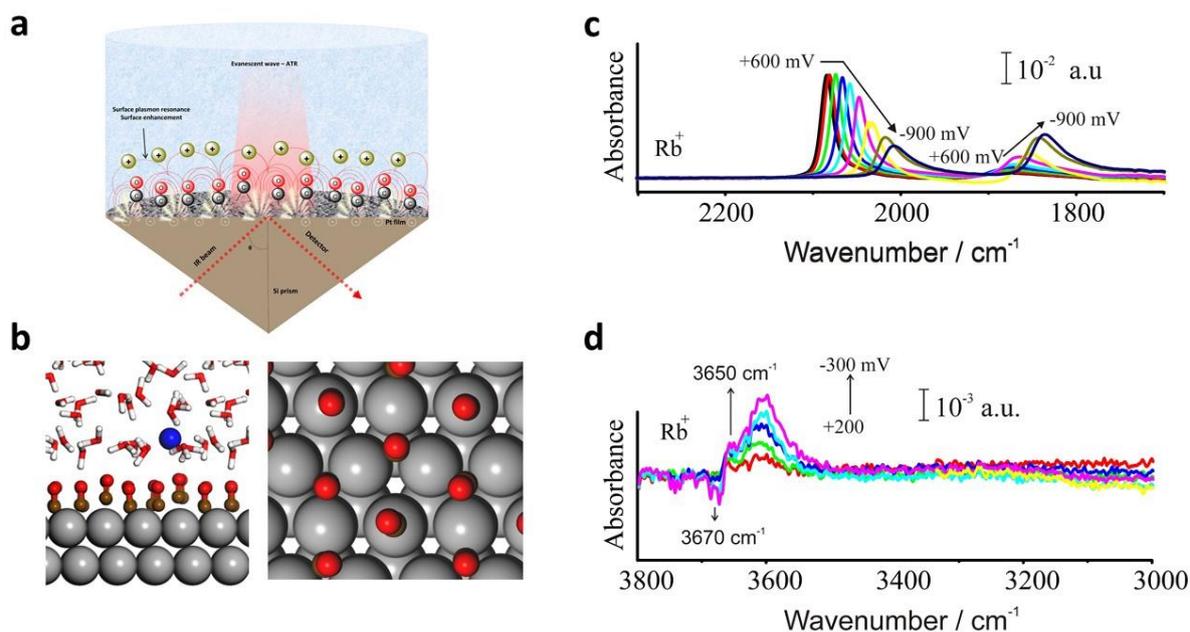
The electric double layers are modelled by inserting Na atoms at  $\sim 3 \text{ \AA}$  away from the surface in the Pt(111)-CO/water interface. As expected, Na atoms are spontaneously oxidized to  $\text{Na}^+$  in liquid water. As the whole interface has to be neutral, a charge identical in magnitude but of opposite sign to that of the  $\text{Na}^+$  ions emerges on the Fermi level of the Pt(111)-CO surface. The charge density ( $\sigma$ ) on the Pt(111)-CO surface is therefore

determined by the number of Na atoms inserted in the simulation cell. In this work, we have modelled three Pt(111)-CO/water interfaces:  $\sigma = 0, -14.6$  and  $-29.2 \mu\text{C cm}^{-2}$ .

The DFT implemented in CP2K is based on a hybrid Gaussian plane wave (GPW) scheme, the orbitals are described by an atom-centred Gaussian-type basis set, and an auxiliary plane wave basis set is used to re-expand the electron density in the reciprocal space. The 2s, 2p electrons of O, 2s, 2p electrons of C, 2s, 2p, 3s electrons of Na, and 5d, 6s electrons of Pt are treated as valence, the rest core electrons are represented by Goedecker-Teter-Hutter (GTH) pseudopotentials. The Gaussian basis set is double- $\zeta$  with one set of polarization functions (DZVP), and the energy cut-off is set to 400 Ry. We employ BLYP functional to describe the exchange-correlation effects, and the dispersion correction is applied in all calculations with the Grimme D3 method. The second-generation Car-Parrinello molecular dynamics is used for the DFTMD simulation, and the target temperature is set to 330 K. The discretized integration step is 0.5 fs, with  $\gamma L = 0.001 \text{ fs}^{-1}$ . The first 2 ps of the simulation trajectory was regarded as the equilibration period, then followed by a production period of 15~20 ps. Due to the large size of the cell, only the  $\Gamma$  point in reciprocal space was used. To obtain the vibrational density of states (VDOS) spectra of CO, the velocity-velocity autocorrelation function is used.

### **3. Results and discussion**

We use ATR-SEIRAS to analyse cation effects on the C-O and O-H stretching modes of  $\text{CO}_{\text{ad}}$  and of interfacial water, respectively, in CO-saturated solutions. ATR-SEIRAS (Fig. 1a) combines the ATR geometry with the short range of the SEIRA effect for improved interfacial sensitivity [29,30]. A deeper insight into the EDL structure was reached with AIMD simulations making use of the computational standard hydrogen electrode (SHE) [31]. Figure 1b illustrates the model of the Pt(111)-CO/water interface used for AIMD simulations.



**Figure 1.** (a) Schematic illustration of the surface sensitivity of surface-enhanced infrared absorption spectroscopy in the attenuated total reflection mode (ATR-SEIRAS), due to the combination of the thin layer of electrolyte probed by the evanescent wave (typical of the ATR configuration) and to the short range of the SEIRA effect generated by surface plasmon excitation within the rough metal film deposited on the Si prismatic window. (b) Model of the Pt(111)-CO/water interface used for AIMD simulations. Pt, C, O, H and cation are coloured by grey, brown, red, white and blue, respectively. The Pt(111) surface is covered by the (2 x 2)-3CO structure ( $\theta = 0.75$  ML) known to exist on this surface in CO-saturated solutions at  $E \leq 0.45$  V vs. RHE [32]. (c) ATR-SEIRA spectra of a Pt electrode in CO-saturated 0.1 M Rb<sub>2</sub>SO<sub>4</sub> in the spectral region corresponding to the C-O stretching of adsorbed CO. The spectra are shown at potential intervals of 0.20 V for the sake of clarity and were calculated using the spectrum of the CO-free Pt surface at the open-circuit potential as background. (d) ATR-SEIRA spectra of a Pt electrode in CO-saturated 0.1 M Rb<sub>2</sub>SO<sub>4</sub> in the spectral region corresponding to the O-H stretching of H<sub>2</sub>O. The spectra are shown at potential intervals of 100 mV for the sake of clarity and were calculated using the spectrum of CO-covered Pt surface at +0.4 V as background.

Experiments were performed with 14 different cations: H<sup>+</sup>, NH<sub>4</sub><sup>+</sup>, Li<sup>+</sup>, Na<sup>+</sup>, K<sup>+</sup>, Rb<sup>+</sup>, Cs<sup>+</sup>, Mg<sup>2+</sup>, Ca<sup>2+</sup>, Sr<sup>2+</sup>, tetramethylammonium (TMA<sup>+</sup>), tetraethylammonium (TEA<sup>+</sup>), tetrapropylammonium (TPA<sup>+</sup>) and tetrabutylammonium (TBA<sup>+</sup>). Figs. 1c and d show the

spectra obtained with  $\text{Rb}^+$  in the CO and OH stretching regions, respectively. Spectra for all the cations can be found in Fig. S1.

**3.1. Cation effect on the potential dependence of  $\bar{\nu}_{\text{CO}_L}$  and on  $C_T$ : EDL model.** In the CO stretching region, all the spectra consist of two bands that red-shift with increasingly negative potential (Fig. 1c and Fig. S1a). The band between ca. 2000 and ca. 2090  $\text{cm}^{-1}$  corresponds to linearly adsorbed CO ( $\text{CO}_L$ ), while the band between 1800 and 1900  $\text{cm}^{-1}$  corresponds to bridge-bonded adsorbed CO ( $\text{CO}_B$ ). The dependence of the  $\text{CO}_L$  frequency ( $\bar{\nu}_{\text{CO}_L}$ ) on potential is shown in Fig. 2a. There is a wide potential region of linear dependence (shown in more detail in Fig. S2). The deviation from linearity at negative potentials is not central to this discussion and is explained in more detail in the Supporting Information (SI). We will focus our discussion on  $\text{CO}_L$ .  $\text{CO}_B$  shows a similar dependence on potential, but the bands are broader, and a detailed analysis is subject to a larger error.

The STR can be expressed as:

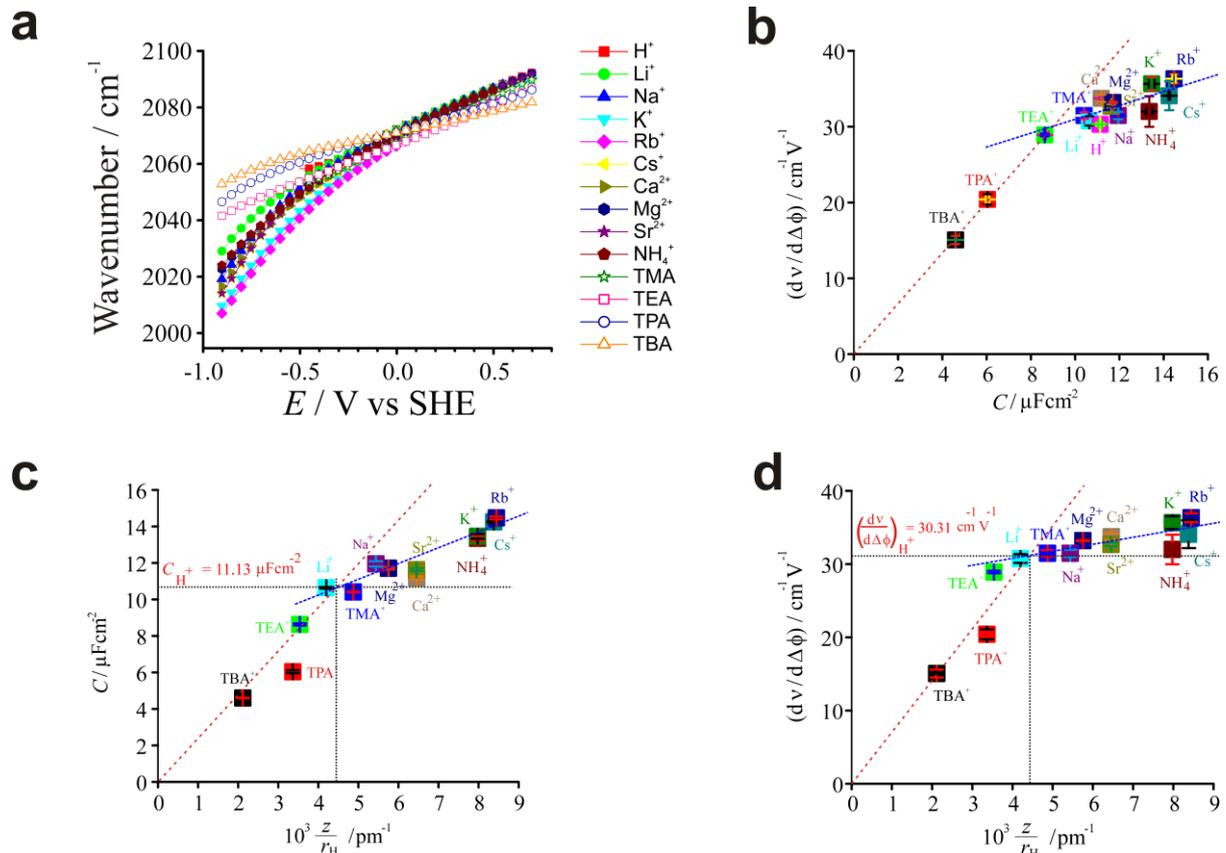
$$\frac{\partial \bar{\nu}_{\text{CO}_L}}{\partial \Delta \phi} = \frac{\partial \bar{\nu}_{\text{CO}_L}}{\partial F} \frac{\partial F}{\partial \Delta \phi} \quad \text{Eq. 1}$$

where, in a first approximation,  $\frac{\partial \bar{\nu}_{\text{CO}_L}}{\partial F}$  can be considered as constant. As the interfacial

electric field is  $F = \frac{\Delta \phi}{\Delta x_{\text{dl}}}$ , where  $\Delta x_{\text{dl}}$  is the thickness of the EDL,  $\frac{\partial F}{\partial \Delta \phi} = \frac{1}{\Delta x_{\text{dl}}}$ . Therefore,

assuming that  $\Delta x_{\text{dl}}$  is determined by the size of the cation, a direct proportionality between the STR and the inverse of the cation radius should be obtained. Fig. 2c shows a plot of the STR (in the region of linear dependence between  $\bar{\nu}_{\text{CO}_L}$  and the potential) vs. the charge number to hydrodynamic radius ratio (*i.e.*, the radius of the cation plus its hydration shell). The latter was calculated from the cation's limiting molar conductivity,  $\lambda$  (see SI). Similar cation effects on the STR have been found by Roth and Weaver [16] for  $\text{CO}_{\text{ad}}$  on Pt in non-aqueous media, and have also been recently reported for  $\text{CO}_{\text{ad}}$  on Cu [15,33]. The values of the STR for all the cations used in this study are provided in Table S1. The value obtained for  $\text{H}^+$  is in

good agreement with literature values for polycrystalline Pt [34]. Although a reasonable direct proportionality is obtained as the cation size decreases from aqueous TBA<sup>+</sup> to aqueous Li<sup>+</sup>, for smaller cations ( $\frac{z}{r_H} > 4.5 \times 10^{-3}$ ) there is a clear deviation from the expected behaviour.



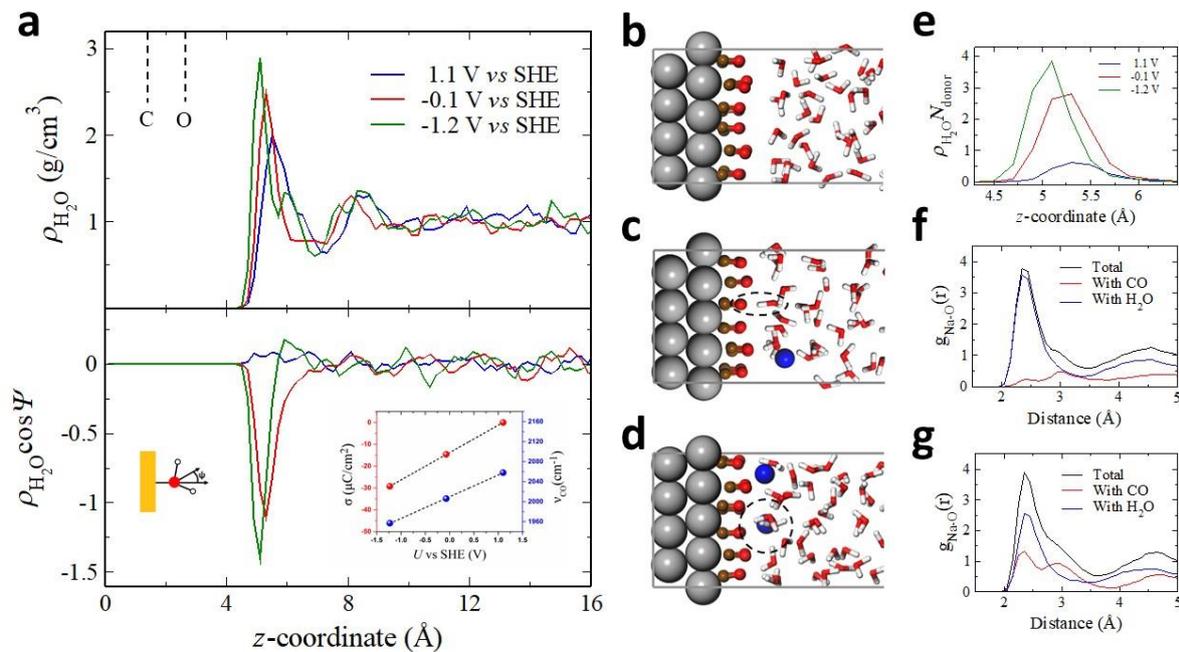
**Figure 2.** (a) Plot of the CO<sub>L</sub> stretching frequency as a function of the electrode potential for all the cations used in this study. (b) Plot of the Stark tuning rate vs. the interfacial capacitance for all 14 cations used in this study. The dashed red line is the best linear fit to the data corresponding to the three largest cations (TBA<sup>+</sup>, TPA<sup>+</sup> and TEA<sup>+</sup>). The dashed blue line is a linear fit to the rest of the data. (c) and (d), Plots of the interfacial capacitance and the Stark tuning rate, respectively, as a function of the ratio between the charge number of the cation and the cation's hydrodynamic radius,  $\frac{z}{r_H}$ , as obtained from the corresponding limiting ionic conductivity,  $\lambda$ . The dashed horizontal lines in c and d correspond to the interfacial capacitance and the Stark tuning rate, respectively, when H<sup>+</sup> is the electrolyte cation, and the vertical dashed lines correspond to the resulting estimated effective radius of hydrated H<sup>+</sup> at the electrical double layer.

A similar deviation from the expected behaviour is observed for the interfacial capacitance, as, in the simplest possible model of the EDL,  $C_H = \frac{\varepsilon\varepsilon_0}{\Delta x_{dl}}$ , with  $\varepsilon_0$  the vacuum permittivity and  $\varepsilon$  the dielectric constant of the interfacial region As shown in Fig. 2d, a reasonable direct proportionality is obtained again only for cations larger than aqueous  $\text{Li}^+$ . The values of  $C_T$  for all the cations used in this study are reported in Table S1.

Please note that, due to proton hopping between water molecules [35], the size of hydrated  $\text{H}^+$  cannot be determined from  $\lambda$ , and  $\text{H}^+$  has therefore not been included in Figs. 2c and d. However, these figures allow for an estimation of the effective size of hydrated  $\text{H}^+$  of approximately 2.2 Å. It is also worth noting that both the STR and  $C_H$  show a good correlation, not with the radius of the cation, but with  $\frac{z}{r_H}$ . In other words, the ratio between the cation's charge and its hydrodynamic radius behaves as the effective cation size. This has to be due to the fact that the potential generated by a central ion decays with the distance from the ion,  $r$ , as  $\frac{z}{r}$  and, consequently, a monovalent ion with a hydrodynamic radius  $r_H$  and a divalent ion with a hydrodynamic radius  $2r_H$  will induce exactly the same potential at the surface of the solvation shell, leading to the same  $C_T$  and the same STR, as found experimentally (Figs. 2c and d). This also implies that the potential drop across the cation's hydration shell contributes to the overall interfacial potential drop and must be taken into account when describing the structure of the EDL.

AIMD simulations offer a deeper insight into the EDL structure. We first calculate the electrode potential at different surface charge densities ( $\sigma$ ). Computing costs limit our calculations to  $\text{Na}^+$  and to  $\sigma = 0, -14.6$  and  $-29.2 \mu\text{C cm}^{-2}$ , corresponding to 0, 1 or 2  $\text{Na}^+$  ions in the simulation cell, respectively. The computed pzc of the Pt(111)-CO/water interface is  $\sim 1.1$  V vs SHE (Fig. 3a, inset), in good agreement with the experimental value [18]. The computed  $C_T$  is  $13 \mu\text{F cm}^{-2}$  (Fig. 3a, inset), also in good agreement with the experimental

result for  $\text{Na}^+$  (Table S1). Computed IR spectra of the  $(2 \times 2)$  structure of CO on Pt(111) can be found in the SI (Fig. S3), and yield a computed STR of  $40 \text{ cm}^{-1} \text{ V}^{-1}$  for  $\text{CO}_L$ , in reasonable agreement with experimental values in this work and elsewhere [20,36,37]. In summary, our AIMD simulation captures well the structure and properties of the EDL. At the pzc, the surface water is approximately  $6 \text{ \AA}$  away from the Pt(111) surface and  $\sim 2 \text{ \AA}$  away from the oxygen atom of  $\text{CO}_{\text{ad}}$ , and shows a negligible degree of dipole orientation (Fig. 3a), in agreement with previous work with this and other surfaces [18,31]. At negative charge densities, water approaches closer to the electrode and orients with H pointing towards the surface (Fig. 3a, c, and d).



**Figure 3.** (a) Water density ( $\rho_{\text{H}_2\text{O}}$ , upper panel) and dipole orientation ( $\rho_{\text{H}_2\text{O}} \cos \Psi$ , lower panel) distributions along the surface normal direction (z-coordinate) at the pzc (+1.1 V), -0.1 V and -1.2 V. Zero in the z-coordinate axis corresponds to the Pt(111) surface, and the averaged positions of the C and O atoms of adsorbed CO have been labelled. The inset shows the surface charge density on the CO-covered Pt(111) electrode ( $\sigma$ ) and the vibrational frequency of  $\text{CO}_L$  ( $\nu_{\text{CO}}$ ) at three different potentials, from which a pzc of +1.1 V and a capacitance of  $13 \mu\text{F cm}^{-2}$  results. (b), (c) and (d) show representative snapshots from AIMD at the pzc (+1.1 V), -0.1 V and -1.2 V, respectively. (e) Number of hydrogen bonds ( $\rho_{\text{H}_2\text{O}} N_{\text{donor}}$ ) formed between CO and water along the z-coordinate at the pzc (+1.1 V), -0.1 V and -1.2 V, averaged in the XY plane. (f) and (g) show the distribution of

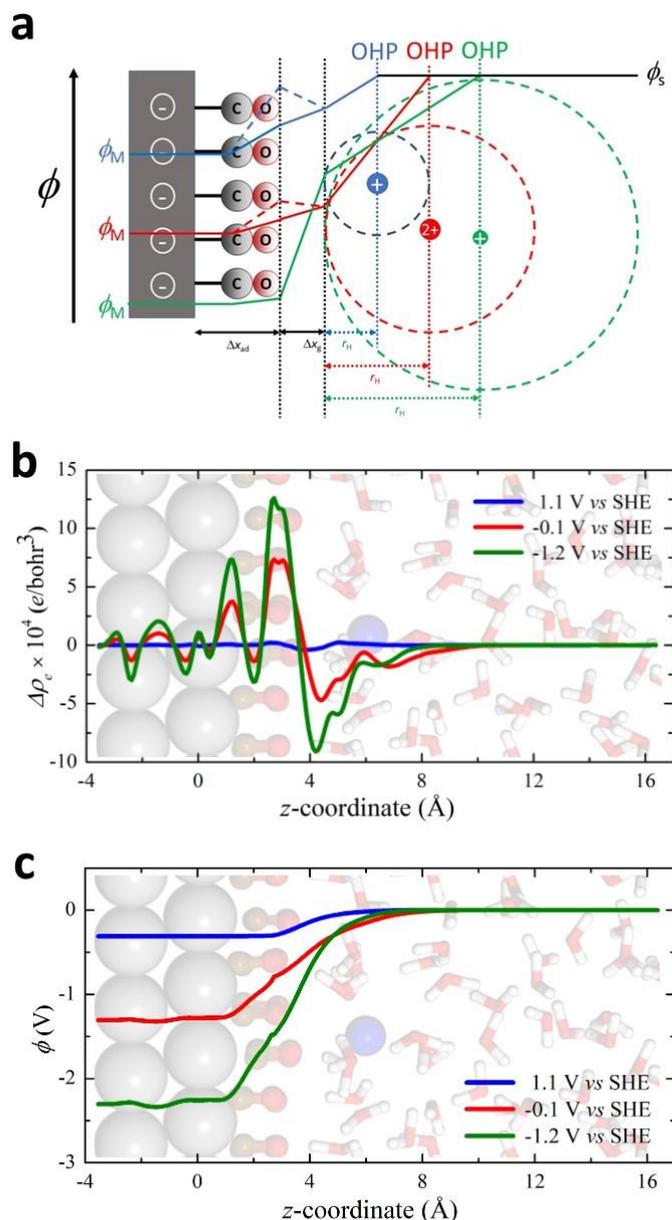
radial distances between Na<sup>+</sup> and O at -0.1 and -1.2 V vs SHE, respectively. The blue and red lines indicate the coordination of Na<sup>+</sup> with H<sub>2</sub>O and CO, respectively, and the black line corresponds to the total number of oxygen atoms coordinated to Na<sup>+</sup>.

Fig. 4 shows a model of the EDL which accounts for the experimental observations (Fig. 4a), the AIMD distribution of charge density (Fig. 4b) and the AIMD potential profile (Fig. 4c). Fig. 4a takes into account that the effective radius of a cation at the interface is given by  $r_{\text{eff}} = \frac{r_{\text{H}}}{z}$ , as suggested by Figs. 2c and d and discussed above. At the pzc, the potential drop across the interface is small, but not zero. At -0.1 and -1.2 V the potential drop has three contributions (Figs. 4a and c), corresponding to (i) the CO adlayer, (ii) the ca. 2 Å wide gap between the electrode surface and interfacial water and (iii) the slice of interfacial water up to the OHP. The accumulation of electronic density on the oxygen atom of CO<sub>ad</sub> at -0.1 and, particularly, -1.2 V, is clear in Fig. 4b. This is equivalent with having three capacitors in series, corresponding to the capacitance of the CO adlayer ( $C_a = \frac{\epsilon_0 \epsilon_a}{\Delta x_a}$ ), the capacitance of the gap ( $C_g = \frac{\epsilon_0 \epsilon_g}{\Delta x_g}$ ) and the capacitance of the water layer between the gap and the OHP ( $C_w = \frac{\epsilon_0 \epsilon_w}{r_{\text{eff}}}$ ), where  $\epsilon_a$ ,  $\epsilon_g$  and  $\epsilon_w$  are the dielectric constants of the CO adlayer, the gap and the water layer, respectively, and  $\Delta x_a$  and  $\Delta x_g$  are the thicknesses of the CO adlayer and the gap, respectively. *I.e.*, within this picture,  $C_T$  is given by:

$$C_T \approx \frac{1}{C_H} = \frac{1}{C_a} + \frac{1}{C_g} + \frac{1}{C_w} = \frac{\epsilon_0 \epsilon_a \epsilon_g \epsilon_w}{\epsilon_g \epsilon_w \Delta x_a + \epsilon_a \epsilon_w \Delta x_g + \epsilon_a \epsilon_g r_{\text{eff}}} \quad \text{Eq. 2}$$

The fraction of the potential drop falling within the CO adlayer will be given by  $\frac{\Delta \phi_a}{\Delta \phi} = \frac{C_T}{C_a}$  (similarly, the fractions of the potential drop corresponding to the gap and the interfacial water layer will be given by  $\frac{\Delta \phi_g}{\Delta \phi} = \frac{C_T}{C_g}$  and  $\frac{\Delta \phi_w}{\Delta \phi} = \frac{C_T}{C_w}$ , respectively). The CO<sub>ad</sub> stretching frequency is determined by the potential drop across the CO adlayer,  $\Delta \phi_a$ , or, equivalently, by the electric field felt by the CO adlayer,  $F_a = \frac{\Delta \phi_a}{\Delta x_a}$ . Accordingly:

$$\frac{\partial \bar{v}_{\text{CO}_L}}{\partial \Delta \phi} = \frac{\partial \bar{v}_{\text{CO}_L}}{\partial F_a} \frac{\partial F_a}{\partial \Delta \phi_a} \frac{\partial \Delta \phi_a}{\partial \Delta \phi} = \frac{\varepsilon_g \varepsilon_w}{\varepsilon_g \varepsilon_w \Delta x_a + \varepsilon_a \varepsilon_w \Delta x_g + \varepsilon_a \varepsilon_g r_{\text{eff}}} \frac{\partial \bar{v}_{\text{CO}_L}}{\partial F} = \frac{1}{\varepsilon_o \varepsilon_a} \frac{\partial \bar{v}_{\text{CO}_L}}{\partial F} C_T \quad \text{Eq. 3}$$



**Figure 4.** (a) Model illustrating the potential profile across the electrode-electrolyte interface, and the effect of the cation size on the interfacial properties and on the polarisation of the CO adlayer. The red cation is divalent, and therefore the situation depicted corresponds to twice the charge density than in the case of the monovalent blue and green cations.  $\Delta x_a$  and  $\Delta x_g$  correspond to the thickness of the CO adlayer and the vacuum gap, respectively, and  $r_H$  is the radius of the hydrated cation. The dashed lines illustrate the potential drops across the CO adlayer and the vacuum gap had  $\varepsilon_a$  remained constant. (b) Distribution of excess electronic density along the surface normal direction, as calculated from AIMD simulation and averaged in the XY plane. The excess

electronic density is defined as  $\Delta\rho_e = \rho_{interface} - \rho_{Pt} - \rho_{water} - \rho_{Na}$ . (c) Potential distribution across the interface. The zero  $z$ -coordinate corresponds to the Pt(111) surface. Blue, red and green curves correspond to the interface at the pzc (+1.1), -0.1 V and -1.2 V, respectively.

The clear direct proportionality between  $C_T$  and  $\frac{z}{r_H}$  for  $\frac{z}{r_H} < 4.5 \times 10^{-3} \text{ pm}^{-1}$  ( $r_{\text{eff}} > 2.2 \text{ \AA}$ ) shown in Fig. 2c suggests that, for the larger cations, the interfacial capacitance is dominated by that of the region between the end of the gap and the OHP, *i.e.*,  $\frac{1}{C_w} \gg \frac{1}{C_a} + \frac{1}{C_g}$  (see Eq. 2). As  $r_{\text{eff}}$  decreases, the contribution of the other two regions to the total capacitance and the overall potential drop become non-negligible, resulting in a deviation from the direct proportionality between  $C_T$  and  $\frac{z}{r_H}$ .

According to Eqs. 2 and 3, exactly the same dependence on  $\frac{z}{r_H}$  should be expected for the STR and  $C_T$ , and the STR should be directly proportional to  $C_T$  if  $\epsilon_a$  remains constant. Such behaviour would imply a rather expectable cation-dependent polarisation of  $\text{CO}_{\text{ad}}$ , sufficient to explain the trend observed with the largest cations. However, the deviation from a direct proportionality with  $\frac{z}{r_H}$  for  $r_{\text{eff}} < 2.2 \text{ \AA}$  is stronger for the STR (Fig. 2c) than for  $C_T$  (Fig. 2d), which leads to the lack of direct proportionality between STR and  $C_T$  for cations smaller than  $\text{TEA}^+$  observed in Fig. 2b. Assuming, in a first approximation, that  $\frac{\partial \bar{v}_{\text{CO}_{\text{L}}}}{\partial F}$  is a constant, the deviation from a direct proportionality is only possible if, for sufficiently small cations,  $\epsilon_a$  becomes cation dependent. *I.e.*, the small distance between the OHP and the electrode surface must provoke too large an electric field across  $\text{CO}_{\text{ad}}$ , to which it reacts with an increase of its polarizability (note that  $\epsilon = 1 + \chi$ , where  $\chi$  is the polarizability), which results in an electric field across  $\text{CO}_{\text{ad}}$  smaller than it would have been had the polarizability not increased. An increase in  $\epsilon_a$  must lead to a stronger deviation from a direct proportionality with  $\frac{z}{r_H}$  for the STR, because, while increasing  $\epsilon_a$  affects both the numerator and the denominator of  $C_T$ , it

affects only the denominator of the STR. Actually, the higher sensitivity of the STR to changes in  $\epsilon_a$  was to be expected, as capacitance measurements probe the whole interface, while  $\frac{\partial \bar{\nu}_{\text{CO}_L}}{\partial \Delta\phi}$  probes only the CO adlayer and must therefore be more sensitive to variations in its properties.

**3.2. Hydrogen bonding between interfacial water and CO<sub>ad</sub>.** At  $E < \text{pzc}$ , the interfacial water layer approaches closer to the surface and orientates itself with H pointing towards the surface (Fig. 3a, c and d). This will favour formation of hydrogen bonds with CO<sub>ad</sub>, as confirmed by AIMD (Fig. 3e). The O-H stretching region of our spectra provides evidence of increased hydrogen bonding between interfacial water and CO<sub>ad</sub> at increasingly negative potentials, as discussed below.

The spectra with H<sup>+</sup>, NH<sub>4</sub><sup>+</sup>, the alkaline-metal cations, Mg<sup>2+</sup>, Ca<sup>2+</sup> and Sr<sup>2+</sup> (see Fig. 1d and Fig. S1b) are consistent with Yamakata's and Osawa's [38–40], who used CO-covered Pt electrodes and ATR-SEIRAS to study the hydration shell of several hydrophilic and hydrophobic cations. The spectra are dominated by positive bands at  $\sim 3600 \text{ cm}^{-1}$ , corresponding to the O-H stretching of water molecules in the hydration shell of hydrophilic cations [38,39]. The lower frequency and broader bands observed in the case of NR<sub>4</sub><sup>+</sup> [38–40] (not shown in Fig. 1Sb) is evidence that hydrogen bonding within their solvation shells is stronger [38,39]. The relatively sharp negative band at  $3670 \text{ cm}^{-1}$  in Fig. 1d and Fig. S1b is characteristic of water with a low degree of hydrogen bonding, and has been assigned to the OH stretching of water molecules at the interface between hydrophobic CO-covered Pt and the electrolyte [41–43]. It is accompanied by a positive band around  $3650 \text{ cm}^{-1}$  that Yamakata and Osawa [39] have attributed to a red-shift of the band at  $3670 \text{ cm}^{-1}$ , due to the approach of the hydrated cations, resulting in a bipolar band that grows with increasing negative potential. Although we agree that this band is indeed due to a red-shift of the  $3670 \text{ cm}^{-1}$  band, we suggest instead that such shift is due to the increased hydrogen-down orientation of

interfacial water (Figs. 3c and d) and the resulting increased degree of hydrogen bonding with  $\text{CO}_{\text{ad}}$  (Fig. 3e).

Figs. 2c and d suggest that the cations' primary hydration shell is retained at the electrochemical double layer, at least at  $E > -0.1$  V. The corresponding band about  $3600\text{ cm}^{-1}$  red-shifts with increasing negative potentials (Fig. S4), providing evidence of increased polarisation of solvation water. (This band and this effect are not observed with  $\text{NR}_4^+$ , for this reason and for the sake of clarity, these cations were not included in Fig. S1b.) However, both its frequency and its shift with potential are cation independent (Fig. S4), *i.e.*, any cation effect on the degree of polarisation or on how much that polarisation increases with increasing negative potential is too small to be detected.

**3.3 Relevance for the  $\text{CO}_2\text{RR}$ .** Although the intense electric field at the EDL has been previously proposed as responsible for the remarkable cation sensitivity of both the activity and selectivity of the  $\text{CO}_2\text{RR}$  [14,44–46], ours is the first direct experimental evidence that field-induced effects go beyond increasing the degree of polarisation of adsorbed species. With sufficiently small cations, the close proximity of the OHP to the electrode surface generates an electric field across  $\text{CO}_{\text{ad}}$  intense enough to lead to a rearrangement of its electronic density large enough to result in an increased polarizability, the degree of which is cation dependent. This must have an effect on the adsorption energy of  $\text{CO}_{\text{ad}}$  (a critical reaction intermediate in the path to both  $\text{C}_1$  and  $\text{C}_2$  hydrogenated products) much stronger than that resulting from a simple increase in its polarisation. Other reaction intermediates, such as adsorbed OCCO and OCCOH, can also be affected by this kind of effects. Altogether, this must have a considerable impact on the apparent activation barriers of the different possible pathways. Although our work has focused on saturated CO adlayers on Pt, there is experimental evidence that the effects reported here are also at work for CO on Cu electrodes [33], probably the most studied material for  $\text{CO}_2$  reduction.

Our spectra (Fig. 1d and Fig. S1b) and those in previous work [38–40] in the O-H stretching region, show evidence of increased hydrogen bonding at increasing negative potentials. Combined with the clear cation dependence of  $C_T$  (Fig. 2c), and because the charge density at a given potential will be larger the larger  $C_T$ , and so will be the degree of dipole orientation, we can conclude that the cation-dependent intensity of the electric field at the EDL must lead to a cation-dependent degree of hydrogen bonding of interfacial water with  $\text{CO}_{\text{ad}}$ , which contributes to the effect of cations on the  $\text{CO}_2\text{RR}$ , as recently suggested [15]. Although we have not found direct evidence in this work of differences in the degree of polarisation of solvation water for different cations, this might simply mean that the effect is too small to be detected spectroscopically. However, previous evidence from our own group [5] is strong enough to suggest that the cation-dependent intensity of the electric field at the EDL leads to a cation-dependent buffering capacity at the interface, which also contributes to the cation effect on the  $\text{CO}_2\text{RR}$ .

#### 4. Conclusions

A detailed analysis of the correlation between  $C_T$  and the STR, and of each one of these magnitudes with  $\frac{z}{r_H}$ , combined with AIMD simulations, allows to formulate a detailed EDL model, in which the size of the hydrated cation determines the location of the OHP, which, for the same potential drop across the interface, results in a cation-dependent intensity of the interfacial electric field. The most relevant consequences of this are:

1. A cation dependent effective potential at the plane where the electron transfer happens, as recognised in early work [8]. This will affect the apparent activation energy of the overall reaction and, through it, its activity. It will also affect the activation energy of each individual step involving an electron transfer, including those determining the rates of parallel paths, and will also have, therefore, an effect on the reaction selectivity and on how it varies with potential.

2. A cation-dependent reorganisation of the electronic density of adsorbates. Depending on the cation and the adsorbate, this might result in a simple cation-dependent polarisation or have a stronger effect and lead to a change in the adsorbate's polarizability, as shown here for the case of  $\text{CO}_{\text{ad}}$  and cations smaller than  $\text{TEA}^+$ . If the electric field-induced reorganisation of electronic density is different for different adsorbed intermediates this will affect their adsorption energy in different degrees and will lead to a change in the apparent activation energy of the different possible paths.
3. As the pzc should, in principle, not be affected by the cation, smaller cations that result in larger  $C_T$  will lead to a larger negative charge density at the same applied potential,  $E$  (as far as  $E < \text{pzc}$ ). This will result in a cation-dependent degree of orientation of interfacial water dipoles and, consequently, to differences in the degree of hydrogen bonding of interfacial water with adsorbed intermediates. In the particular case of  $\text{CO}_{\text{ad}}$ , this will, furthermore, be enhanced by the accumulation of electronic density on the oxygen atom resulting from the reorganisation of electronic density described in the preceding paragraph. As hydrogen bonding to adsorbed intermediates can play an important role in hydrogenation reactions, this must play a role in determining the activity and selectivity of these reactions.
4. The more intense interfacial electric fields with smaller cations will also result in a higher polarisation of their solvation shells, which will lead to differences in the  $\text{p}K_{\text{a}}$  of cation hydrolysis at the interface. In good agreement with this assertion, recently reported results [5] show that precisely those cations which have been shown here to have a smaller effective radius in aqueous electrolytes have a higher buffering capacity at the interface during the  $\text{CO}_2\text{RR}$  in  $\text{CO}_2$ -bicarbonate buffer solutions.

All these effects contribute to the sensitivity of the activity and selectivity of the  $\text{CO}_2\text{RR}$  to the electrolyte cation and reinforce each other, providing a comprehensive explanation to

previous results [4,5,14,15]. The question is, therefore, not which of them is responsible for the observed trends, but to which extent each one of them is determining those trends, and how they can be controlled to improve the activity and yield the desired product distribution.

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